

# First-principles study of ammonium ions and their hydration in montmorillonites

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Received: 21 November 2012 / Accepted: 21 December 2012 / Published online: 17 January 2013  
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**Abstract** Density functional theory calculations were performed to investigate the adsorption and hydration of an ammonium ion ( $\text{NH}_4^+$ ) confined in the interlayer space of montmorillonites (MMT).  $\text{NH}_4^+$  is trapped in the six-oxygen-ring on the internal surface and forms a strong binding with the surface O atoms. The hydration of  $\text{NH}_4^+$  is affected significantly by the surface. Water molecules prefer the surface sites, and do not bind with the  $\text{NH}_4^+$  unless enough water molecules are supplied. Moreover, the water molecules involved in  $\text{NH}_4^+$  hydration tend to bind with the surface simultaneously. The hydration energy increases with the intercalated water molecules, in contrast to that in gas phase. In addition, the hydration leads to the extension of MMT basal spacing.

**Keywords** Adsorption · Ammonium ions · Density functional theory · Hydration · Montmorillonite

## Introduction

Montmorillonites (MMT) are characterized by their laminar structures consisting of alternative octahedral aluminas and tetrahedral silicas. In a typical MMT structure, isomorphic substitutions of low-valence cations in the silica or the alumina sheet leave extra negative charge in the layer. In

order to balance the negative charge, counterions including  $\text{Na}^+$ ,  $\text{K}^+$ ,  $\text{NH}_4^+$ ,  $\text{Mg}^{2+}$ ,  $\text{Ca}^{2+}$ ,  $\text{Fe}^{3+}$ , etc. intercalate into the interlayer space, resulting in various specific properties like cation exchange, swelling, nano-sheets, etc. [1–10]. The study of  $\text{NH}_4^+$  intercalation has particular importance in petroleum engineering, environment protection, and material research. Ammonium ions are one of the main components of drilling fluid, effective in filtration reduction, swelling inhibition, and well-bore stabilization [11–13]. Ammonium ions also exist in waste water, depositing beneath stratum together with other wastes [11, 14–16]. Since MMT layers are about 1 nm in width, they are natural additives in the preparation of nano-composites [17–23]. Ammonium ions and their hydration are often used to split MMT into separate sheets in nano size. In all these processes, the  $\text{NH}_4^+$ -MMT interaction plays a key role and has attracted much interest in past years. X-ray diffraction (XRD), infrared (IR) spectrum, and scanning electron microscope (SEM) techniques have been employed to investigate the structures of ammonium-clay systems, confirming that  $\text{NH}_4^+$  and its hydrates are confined in the interlayer space and interact with the internal surface [24–30]. To our knowledge, however, neither the adsorption nor the hydration of  $\text{NH}_4^+$  on the internal surface of clays has been theoretically studied although first-principles calculations have been proven an effective way in the study of molecular adsorption on surfaces [31–35]. In this work, we study the structures of  $\text{NH}_4^+$  and its hydrates inside MMT, as well as their interaction with the atoms on the internal surfaces by means of density functional theory (DFT) calculations. Our calculations aim to reveal how an  $\text{NH}_4^+$  ion adsorbs on MMT surface, how water molecules affect the  $\text{NH}_4^+$  adsorption, and how the adsorption affects MMT structures, all of which are interesting for understanding ammonium ions' diffusion, deposition, transportation, and activity inside MMT.

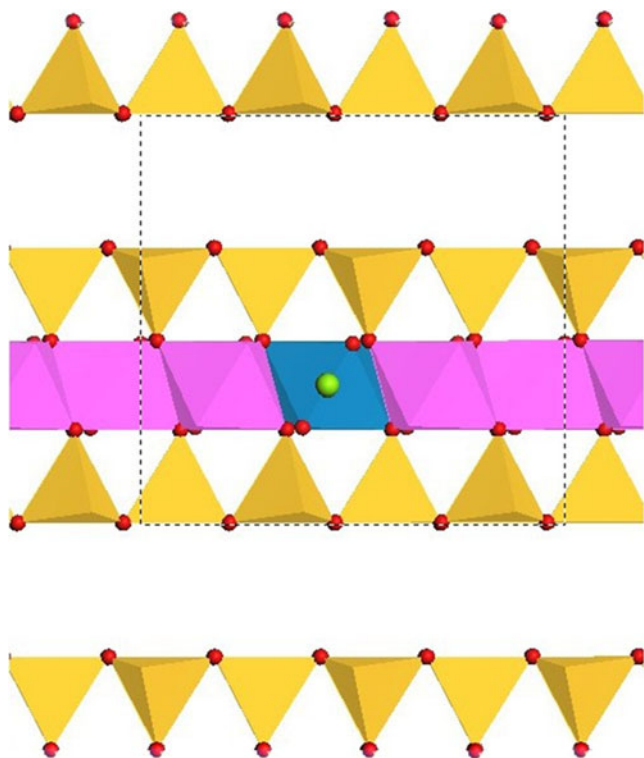
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## Methodology

Several structural models have been reported so far for Li-, Na-, K-, Cs- and Ca-MMT [36–40]. In this work, the initial atomic coordinates of MMT was taken from the measurement of Na- and Ca-MMT by Viani [39] with XRD. In this model, no isomorphous substitution was imposed. To obtain negatively charged layers, the structures with  $\text{Al}^{3+}/\text{Mg}^{2+}$  substitution in the alumina layer was constructed. A  $2 \times 1 \times 1$  supercell with one  $\text{Al}^{3+}/\text{Mg}^{2+}$  isomorphous substitution was used in the calculations. Figure 1 displays the laminar structure of MMT in which two internal surfaces are opposite and form a basal spacing for the encapsulation of counterions and other guest molecules.  $\text{NH}_4^+$  and  $\text{H}_2\text{O}$  were placed randomly in the space, followed by a Monte-Carlo (MC) simulation with CLAYFF force field [41] to locate their optimal adsorption sites. The CLAYFF parameters have been found feasible for clay systems [42–45]. At least two independent MC runs for each configuration were carried out and three or more MC-predicted structures along with some additional structures designed from chemical intuition for each system were collected for further studies at DFT level.

All the DFT calculations were carried out under the generalized gradient approximation (GGA) of Perdew-Burke-Ernzerhof (PBE) parameterization [46], as implemented in the SIESTA program [47, 48]. The non-local

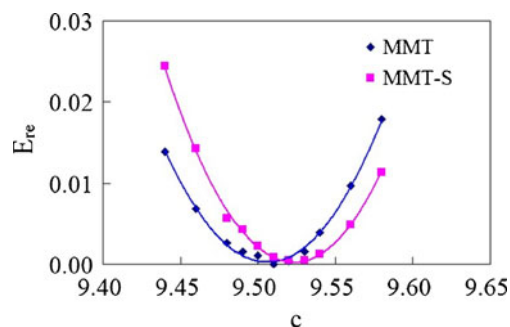


**Fig. 1** The laminar structure of MMT. The tetrahedral silica sheet is in yellow and the octahedral alumina sheet in pink except the blue one in which the  $\text{Al}^{3+}$  is replaced by  $\text{Mg}^{2+}$  (green ball)

norm-conserving pseudopotentials in the Troullier-Martins form [49] were used to describe the core electrons, while the valence wave functions are expanded on (pseudo) atomic orbitals including multiple- $\zeta$  and polarization functions. The standard DZP basis set with orbital-confining cutoff radii of 0.02 Ry and a real-space integration grid with a plane wave cutoff of 220 Ry were used in this work. Calculations were restricted to the  $\Gamma$ -point in the Brillouin zone [50, 51]. As hydrogen bonds (HBs) are formed among  $\text{NH}_4^+$ ,  $\text{H}_2\text{O}$  and the surface, a dispersion potential of the Grimme type [52] was used to deal with the weak interaction in the systems. The optimal basal spacing of each system was determined by optimizing all the atomic positions at a series of spacings. The conjugated gradient approach was used in the geometry relaxation and the convergence criterion was set to 0.01 eV  $\text{\AA}^{-1}$  in atomic force.

## Results and discussion

Viani's model gives only a general atomic arrangement in MMT, whereas the MMT structures vary with the in-layer isomorphous substitution ( $\text{Si}^{4+}/\text{Al}^{3+}$  and/or  $\text{Al}^{3+}/\text{Mg}^{2+}$ ), counterion (charge and ionic radius) intercalation, and ionic hydration, etc. We first compared the MMT structures with and without  $\text{Al}^{3+}/\text{Mg}^{2+}$  substitution in the alumina layer in the absence of counterion or water in the interlayer space. Figure 2 displays their energy variation with the width of basal space, which is the distance between adjacent layers. In the cubic MMT cell, the basal spacing is just the cell parameter along the  $c$ -axis. A small width extension of 0.01  $\text{\AA}$  is noted for the  $\text{Al}^{3+}/\text{Mg}^{2+}$  substitution. Since the substitution leaves an extra negative charge, the internal surfaces become more negative, resulting in stronger electrostatic repulsion between two opposite surfaces. In our calculations, the substitution brings the surface 0.06 e more negative charge, which was obtained by summing up the net charge on all the surface atoms from Mulliken analysis. Since the substitution occurs in the middle alumina layer, such effect is

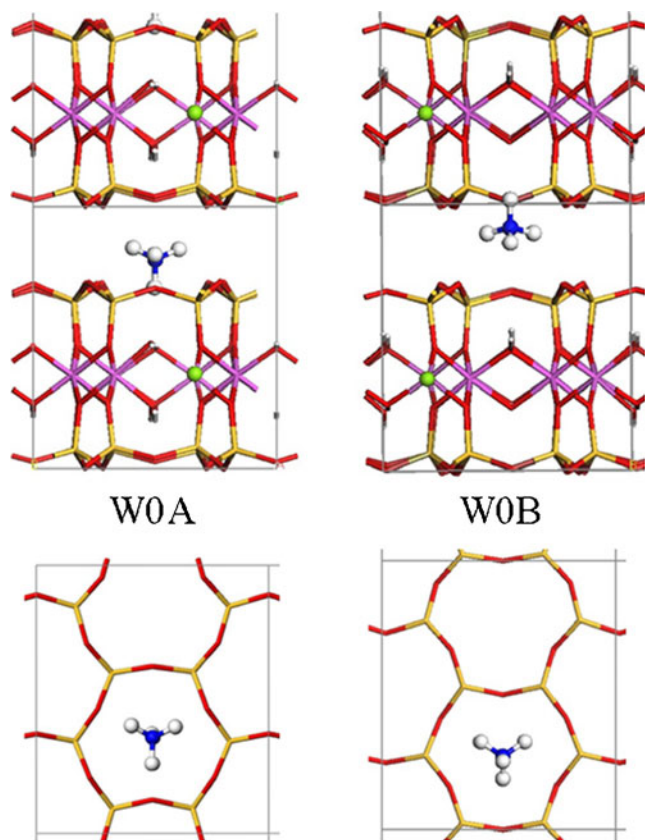


**Fig. 2** Energy variations (in eV) of non-substituted (diamonds) and  $\text{Al}^{3+}/\text{Mg}^{2+}$  substituted (squares) MMT with the basal spacing  $c$  (in  $\text{\AA}$ ). A small increase of  $c$  is noted upon  $\text{Al}^{3+}/\text{Mg}^{2+}$  substitution

rather small. Only the MMT structures with  $\text{Al}^{3+}/\text{Mg}^{2+}$  substitution were studied below. Because of the narrow interlayer space, the guest ions or molecules may interact with both the upper and the lower surfaces simultaneously.

Next, we look at the atomic arrangement on the internal surface, which is covered by O atoms of silicas. Every six O atoms and six Si atoms constitute a cyclic structure in which the O atoms form a six-O-ring (SOR) on the surface. The SOR is a cation trapper with net negative charge. Precisely, three of the six O atoms protrude outward a little more than the other three, having greater tendency to bind with the counterions.

Screened from a number of candidate structures, only two conformers of  $\text{NH}_4^+$  adsorption were identified at DFT level, as presented in Fig. 3, namely W0A and W0B, where Wx denotes the number of water molecules and the last character orders the relative stability. The imagined conformation, two H atoms of  $\text{NH}_4^+$  binding with two surface O atoms, is unstable and changes into either W0A or W0B after geometry optimization. In both structures,  $\text{NH}_4^+$  is trapped by the SOR with one of its H atoms sunk into the surface and forming three hydrogen bonds (HB) with the surface O atoms, while the left four atoms are above the surface. Table 1 lists the computed HB lengths between the adsorbates and between the adsorbates and MMT



**Fig. 3** Side (upper) and top (lower) views of water-free  $\text{NH}_4^+$  inside MMT. Green balls denote  $\text{Mg}^{2+}$  ions that replace  $\text{Al}^{3+}$  in the middle alumina sheet. Blue and gray balls denote N and H atoms respectively

**Table 1** (Averaged) lengths of hydrogen bonds (in Å) between ammonium (A), water (W), and MMT surface (S) atoms in the  $\text{NH}_4^+ \cdot x\text{H}_2\text{O}$ -MMT systems. In parentheses are numbers of hydrogen bonds

	$R_{A-W}$	$R_{A-S}$	$R_{W-W}$	$R_{W-S}$
W0A		1.923 (3)		
W0B		2.089 (3)		
W1A		1.905 (3)		1.715 (2)
W1B		1.905 (3)		1.719 (2)
W1C	1.900 (1)	1.981 (2)		1.810 (2)
W2A	1.887 (1)	1.960 (2)		1.756 (3)
W2B	1.902 (1)	1.815 (3)	1.552 (1)	1.743 (2)
W2C		1.910 (3)	1.531 (1)	1.837 (3)
W2D	1.699 (2)			1.913 (3)
W3A	1.830 (1)	2.022 (2)	1.569 (1)	1.873 (3)
W3B	1.748 (2)	1.907 (2)	1.929 (2)	1.871 (4)
W4A	1.775 (1)	1.935 (3)	1.623 (3)	1.866 (4)
W4B	1.799 (3)		1.919 (2)	1.965 (6)
W5A	1.697 (2)		1.755 (4)	1.913 (6)
W5B	1.788 (3)		1.843 (3)	1.935 (5)

surface. The HB lengths are about 1.923 Å for W0A and 2.089 Å for W0B. These two structures are distinguished by the orientation of three outward N-H bonds, which leads to an energy difference of 0.15 eV, as presented in Table 2.

When one water molecule enters into the interlayer space, three kinds of conformers were identified, as presented in Fig. 4. It is interesting to note that the incoming  $\text{H}_2\text{O}$  binds with the surface sites instead of the  $\text{NH}_4^+$ . It binds solely with either lower (W1A) or upper (W1B) surface via two HBs. W1C is 0.07 eV less stable than W1A, although in W1C  $\text{H}_2\text{O}$  forms one more HB with  $\text{NH}_4^+$ . It is clear that the stability of  $\text{NH}_4^+ \cdot \text{H}_2\text{O}$ -MMT systems depends on not only the HB number but also the HB strength. Because the orientation of  $\text{H}_2\text{O}$  is restricted by the directionality of multiple HBs, the HBs are lengthened and weakened in W1C, as presented in Table 1. Our calculations reveal that water molecules prefer the surface sites to the intercalated  $\text{NH}_4^+$  if enough space is provided.

The second  $\text{H}_2\text{O}$  generates more possible conformers, among which four types of structures were identified, as shown in Fig. 5. In W2A, one of the two water molecules binds with the surface solely, like the  $\text{H}_2\text{O}$  in W1A. The other one binds with both  $\text{NH}_4^+$  and the surface, like the  $\text{H}_2\text{O}$  in W1C. In W2B, one HB is formed between the two water molecules, one of which binds with surface via two HBs and the other binds with both the surface and the  $\text{NH}_4^+$ . Both  $\text{H}_2\text{O}$  form three HBs in total. W2A and W2B have close energy ( $\Delta E=0.02$  eV). W2C is 0.14 eV less stable than W2A. The two water molecules in W2C form an HB between them but do not bind with  $\text{NH}_4^+$ . In W2D, both  $\text{H}_2\text{O}$  form HBs with the  $\text{NH}_4^+$  and the surface, but it is

**Table 2** Basal spacings ( $c$ , in Å), adsorption energy ( $E_{ad}$ , in eV), hydration energy ( $E_H$ , in eV), relative energy ( $E_{re}$ , in eV) surface charge ( $Q(S)$ , in a.u.), net charge on  $NH_4^+$  ( $Q(NH_4)$ , in a.u.) and averaged net charge on water molecules ( $Q(H_2O)$ , in a.u.) of  $NH_4^+ \cdot xH_2O$ -MMT systems

	$c$	$E_{ad}/x$	$E_H$	$E_{re}$	$Q(S)$	$Q(NH_4)$	$Q(H_2O)$
W0A	9.66	4.27		0.00	-1.03	0.53	
W0B	9.64	4.12		0.15	-1.20	0.63	
W1A	9.76	5.52	1.25	0.00	-1.12	0.51	0.06
W1B	9.76	5.46	1.18	0.06	-1.20	0.48	0.05
W1C	9.90	5.45	1.17	0.07	-1.12	0.51	0.06
W2A	10.03	5.55	1.31	0.00	-1.23	0.50	0.15
W2B	10.33	5.54	1.29	0.02	-1.20	0.47	0.12
W2C	10.15	5.48	1.24	0.14	-1.19	0.55	0.14
W2D	10.14	5.33	0.94	0.43	-1.16	0.42	0.13
W3A	10.33	5.65	1.57	0.00	-1.26	0.47	0.13
W3B	11.02	5.62	1.49	0.08	-1.21	0.39	0.09
W4A	10.75	5.70	1.58	0.00	-1.29	0.48	0.07
W4B	11.05	5.68	1.49	0.08	-1.24	0.35	0.09
W5A	11.06	5.76	1.72	0.00	-1.32	0.39	0.06
W5B	11.08	5.75	1.71	0.01	-1.28	0.34	0.08

0.43 eV above W2A. Our calculations reveal that the arrangement of water-water-ammonium (WWA) is energetically favorable on the MMT surface, while the arrangement of water-ammonium-water (WAW), which maximizes the hydration of ammonium ion, is unfavorable on the surface.

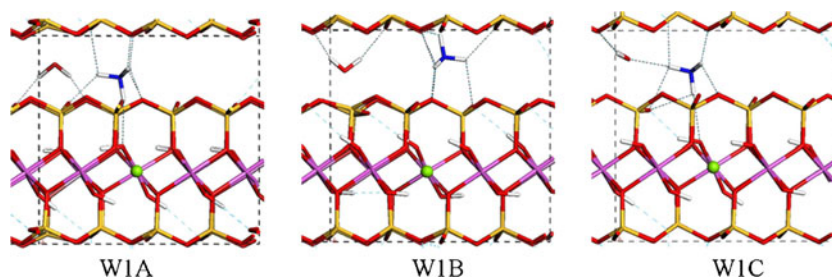
With increasing number of water molecules, the candidate pool of possible conformers becomes very large, making the computations extremely demanding. Meanwhile, the structure of water standing solely on the surface becomes unstable because of the narrow interlayer space. While we are interested in the structural variations caused by  $NH_4^+$  adsorption and its hydration, which are mainly related to the number of water molecules, it is reasonable to impose the above rules of water-ammonium arrangement found in  $NH_4^+ \cdot 2H_2O$ -MMT into the construction of candidate structures. Based on such simplifications, we obtained the low-lying structures of  $NH_4^+ \cdot xH_2O$ -MMT ( $x=3, 4$  and  $5$ ), the best two of which are shown in Fig. 6. In W3A, the two  $H_2O$  form HBs with each other, all three  $H_2O$  form HBs with each other in W3B, while one  $H_2O$  in W3A forms an HB with  $NH_4^+$  and two  $H_2O$  in W3B form HBs with  $NH_4^+$ . Dragged by the water molecules, the H atom of  $NH_4^+$  is no longer below the surface and the  $NH_4^+$  is in fact suspended between two opposite layers. All the adsorbates, water and  $NH_4^+$ , form HBs with the surface. The same structural

motifs are noted in W4A, W4B, W5A and W5B in which the adsorbates ( $H_2O$  and  $NH_4^+$ ) and MMT surfaces are connected via an HB network. With increasing water number,  $NH_4^+$  forms more HBs with water. Although a water molecule binds first with the surface O atoms, then with other water molecules, the volume effect becomes significant when water number increases, making the water molecules bind with  $NH_4^+$  and with each other. From W1 to W5, the number of ammonium-water (A-W) HBs increases from 0 to 3.

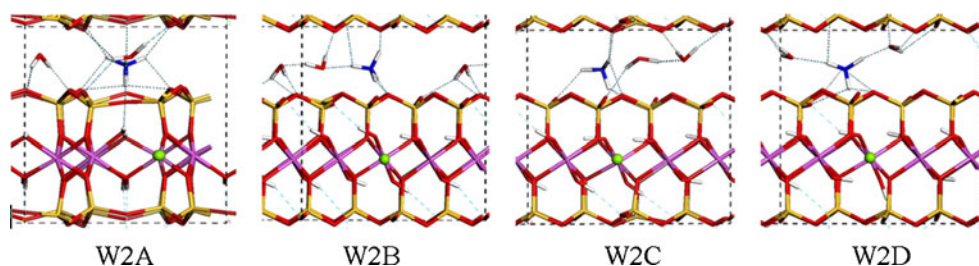
The intercalation of  $NH_4^+$  and water leads to variations in the cell structures and electronic properties. With increasing  $H_2O$ , the basal spacing,  $c$ , changes from 9.66 Å to 11.08 Å (Table 2), implying that the interlayer space is enlarged by incoming  $H_2O$ . Earlier Monte-Carlo simulations [9, 53] pointed out that in Na-MMT,  $c$  increases from 9.7 Å to 12.0 Å when the internal surface is covered by one-layer water molecules. As shown in Figs. 3, 4, 5 and 6, the  $NH_4^+$  and water molecules are basically lined up in the middle of interlayer space. However, there remain some vacancies on the surface for additional water molecules. It is expected that the basal spacing could increase further when more water molecules are adsorbed.

The interaction between adsorbates and surface can be evaluated by adsorption energy, which is defined by

**Fig. 4** Side view of the low-lying  $NH_4^+ \cdot 1H_2O$ -MMT structures in which one ammonium ion and one water molecule adsorb on the MMT internal surfaces



**Fig. 5** Side view of the low-lying  $\text{NH}_4^+ \cdot 2\text{H}_2\text{O}$ -MMT structures in which one ammonium ion and two water molecules adsorb on the MMT internal surfaces

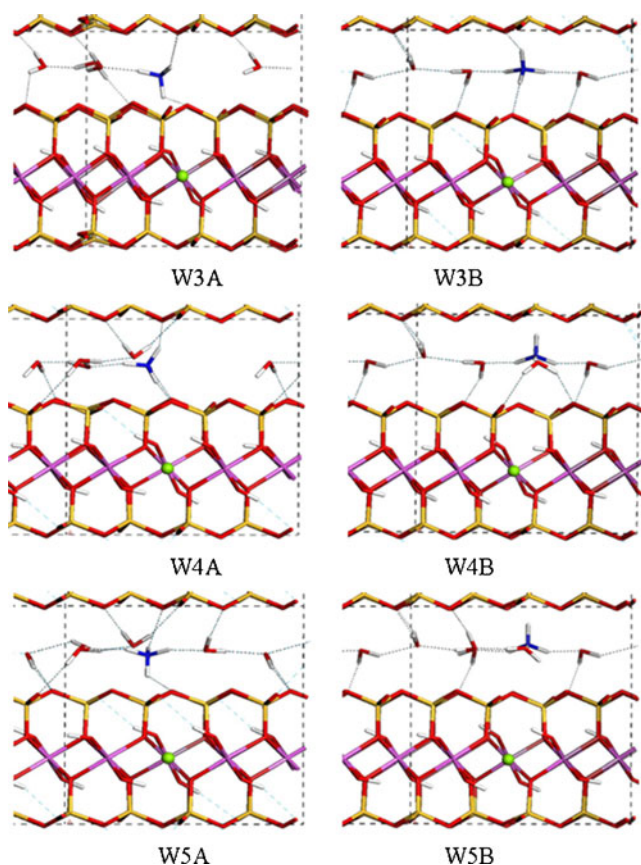


$$E_{\text{ad}} = E(\text{NH}_4^+) + xE(\text{H}_2\text{O}) + E(\text{MMT}) - E(\text{NH}_4^+ \cdot x\text{H}_2\text{O} \cdot \text{MMT}) \quad (1)$$

where  $E(X)$  is the energy of system  $X$ . For an explicit understanding to the hydration effect of the confined  $\text{NH}_4^+$ , we further define hydration energy as

$$E_{\text{H}} = E\{\text{NH}_4^+ \cdot (x-1)\text{H}_2\text{O} \cdot \text{MMT}\} + E(\text{H}_2\text{O}) - E(\text{NH}_4^+ \cdot x\text{H}_2\text{O} \cdot \text{MMT}) \quad (2)$$

The  $E_{\text{ad}}/x$  and  $E_{\text{H}}$  values are given in Table 2 and Fig. 7.  $E_{\text{ad}}/x$  has a sharp increase from  $x=0$  to 1, and then has small

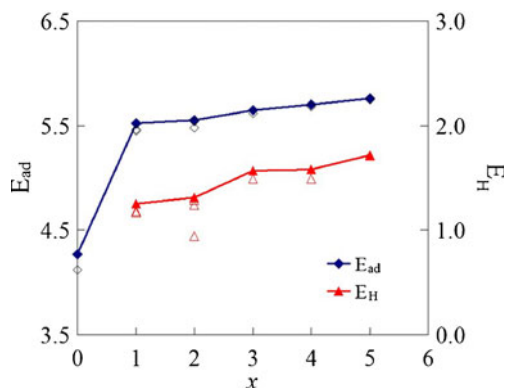


**Fig. 6** Side view of the low-lying  $\text{NH}_4^+ \cdot x\text{H}_2\text{O}$ -MMT ( $x=3, 4$  and  $5$ ) structures in which one ammonium ion and  $x$  water molecules adsorb on the MMT internal surfaces

increases with  $x$ , while  $E_{\text{H}}$  increases steadily from  $x=1$  to 5. The variation of  $E_{\text{ad}}/x$  and  $E_{\text{H}}$  reflect that with increasing  $x$ , more HBs are formed among  $\text{NH}_4^+$ ,  $\text{H}_2\text{O}$ , and MMT surface. The hydration of  $\text{NH}_4^+$  in MMT is an energetically favorable process.

The net charge of the adsorbates and the surface is also an indicator of adsorbate-surface interaction. The net charge of  $\text{NH}_4^+$ ,  $\text{H}_2\text{O}$  and MMT are listed in Table 2. In the water-free W0A structure, about 0.53 e electron moves from surface O to  $\text{NH}_4^+$ . When water enters,  $\text{NH}_4^+$  accepts negative charge from water molecules and becomes less positive. Meanwhile, water accepts electrons from the surface.

$\text{NH}_4^+$  in MMT exhibits different hydration behaviors from in gas phase. In a DFT study, Brugué et al. [54] identified the structures of hydrated  $\text{NH}_4^+$  and found that four water molecules bind with  $\text{NH}_4^+$  one by one via HBs, and the fifth  $\text{H}_2\text{O}$  binds with one of the adsorbed  $\text{H}_2\text{O}$ . These adsorption patterns are not noted among the low-lying structures of hydrated  $\text{NH}_4^+$  in MMT because of the strong binding of surface SOR with  $\text{NH}_4^+$ , as well as the preference of water toward the surface. The BLYP predicted hydration energy in gas is 0.88 eV for first water, and drops to 0.42 eV for the fifth water [54]. In our calculations, the hydration energy increases with the incoming water molecules because of the strong  $\text{H}_2\text{O}$ -surface and  $\text{NH}_4^+$ -surface interaction. The computed hydration energy is 1.25 eV for the first water and increases to 1.72 eV for the fifth.



**Fig. 7** Variations of adsorption energy ( $E_{\text{ad}}$ , in eV) and hydration energy ( $E_{\text{H}}$ , in eV) with the number of water molecules. The values for the most stable configurations are connected with line

## Conclusions

The adsorption and hydration of  $\text{NH}_4^+$  in the interlayer space of montmorillonites (MMT) were investigated using periodic density functional theory calculations with the PBE functional plus dispersion correction. The  $\text{Al}^{3+}/\text{Mg}^{2+}$  substitution in the middle alumina layer makes the internal surfaces more negative and favors the intercalation of counterions.  $\text{NH}_4^+$  is trapped tightly by the SOR on the surface. In its optimal conformer, one of the H atoms is sunk under the surface, while the other three H atoms stretch outward. Different from the cases in gas phase, water molecules prefer the surface sites to the  $\text{NH}_4^+$ . With the addition of water molecules, an HB network is gradually formed between water and surface, water and water, water to  $\text{NH}_4^+$ , and  $\text{NH}_4^+$  to surface. All the incoming water molecules bind with the surface via HBs regardless of their interaction with  $\text{NH}_4^+$ , indicating that the hydration of  $\text{NH}_4^+$  in MMT is highly affected by the surface activity. The interaction between the surface and the adsorbates is evaluated in terms of adsorption energy, hydration energy, and the charge transfer among the groups. A strong binding of  $\text{NH}_4^+$  with the surface is noted, while water molecules bind weakly with the surface and the  $\text{NH}_4^+$ . With increasing number of water molecules, the hydration energy increases, accompanied by the increase of basal spacing.

**Acknowledgments** This work was supported by Open Research Fund of State Key Laboratory of Oil and Gas Reservoir Geology and Exploration (Southwest Petroleum University, No. PLN1118) and National Natural Science Foundation of China (No. 51134004 and 51204142). Part of computations was carried out on the Computer Clusters in Institute of Atomic and Molecular Physics, Sichuan University.

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